

Active Tetrahedral Iron Sites of γ -Fe₂O₃ Catalyzing NO Reduction by NH₃

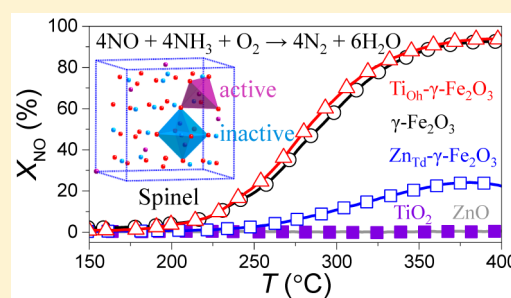
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Supporting Information

ABSTRACT: Maghemite (γ -Fe₂O₃) with a spinel structure, consisting of tetrahedral Fe³⁺ (Fe³⁺_{Td}) and octahedral Fe³⁺ (Fe³⁺_{Oh}) sites, has been intensively investigated as an environmentally benign catalyst for selective catalytic reduction (SCR) of NO_x with NH₃. In most cases, Fe³⁺_{Oh} sites were regarded as catalytically active sites (CASs). Here we identify the CASs in SCR by substituting Fe³⁺_{Oh} or Fe³⁺_{Td} sites of γ -Fe₂O₃ with catalytically inactive Ti⁴⁺ or Zn²⁺, respectively. The SCR activity of γ -Fe₂O₃ is preserved after Ti⁴⁺ doping but drastically decreases when the catalyst is doped with Zn²⁺, demonstrating that Fe³⁺_{Td} sites serve as CASs in SCR. Synchrotron X-ray absorption spectra coupled with density functional theory calculations reveal that the transfer of an electron from inactive Fe²⁺ to active Fe³⁺ in the tetrahedral site is easier than that in the octahedral site, making the tetrahedral iron sites active in SCR.



INTRODUCTION

Selective catalytic reduction (SCR) of NO_x with NH₃ is one of the most promising technologies for controlling NO_x emissions that contribute greatly to atmospheric pollution.^{1,2} Although V₂O₅-based catalysts have been widely applied for SCR in coal-fired power plants,^{3,4} it is necessary to develop vanadium-free SCR catalysts because of the toxicity of vanadium. Fe₂O₃ is an appealing alternative because of its strong resistance to SO₂.^{5–8} There are two types of Fe₂O₃, i.e., hematite (α -Fe₂O₃) and maghemite (γ -Fe₂O₃).⁹ α -Fe₂O₃ has high thermal stability but shows insufficient activity in the operating temperature window of 300–400 °C. On the contrary, γ -Fe₂O₃ is more active than α -Fe₂O₃ at relatively low temperatures,¹⁰ but γ -Fe₂O₃ transforms into α -Fe₂O₃ above 320 °C,¹¹ leading to decreased catalytic activity.

An efficient approach to enhancing the thermal stability of γ -Fe₂O₃ is to substitute part Fe³⁺ with other metal ions.¹¹ To choose desired doping metal ions, identification of catalytically active sites (CASs) of γ -Fe₂O₃ in SCR is one of the important prerequisites. γ -Fe₂O₃ with a spinel structure consists of two types of Fe³⁺: one-third of Fe³⁺ in the tetrahedral sites (Fe³⁺_{Td}) and two-thirds of Fe³⁺ in the octahedral sites (Fe³⁺_{Oh}).¹² It has been overwhelmingly believed that Fe³⁺_{Oh} sites are CASs for two reasons. (i) The mean length (2.09 Å) of the Fe³⁺_{Oh}-O bond is greater than that (1.87 Å) of the Fe³⁺_{Td}-O bond;⁵ thus, the former would be more reducible and reactive. (ii) For metal oxide spinels, the octahedral sites are preferentially located at the outermost surface layers as compared with the tetrahedral sites.¹³ Even though Ti-doped γ -Fe₂O₃ and Zn-doped γ -Fe₂O₃

have been developed for SCR,^{7,14} the local structures of Ti⁴⁺ and Zn²⁺ were not precisely studied, and the authors of these studies still regarded Fe³⁺_{Oh} sites as CASs in SCR.

In this work, we identified CASs of γ -Fe₂O₃ in SCR by using a metal ion doping strategy and chose catalytically inactive Ti⁴⁺ (d⁰) and Zn²⁺ (d¹⁰) to replace Fe³⁺_{Oh} and Fe³⁺_{Td}, respectively. First, the crystal structures of the samples were studied by synchrotron X-ray diffraction (SXRD), and the immediate structures of Ti⁴⁺ and Zn²⁺ were determined by X-ray absorption near-edge structure (XANES) and extended X-ray absorption fine structure (EXAFS) spectra. Second, the electronic structures of the Fe 3d orbitals were characterized by Fe L₃-edge X-ray absorption spectra coupled with density functional theory calculations. Finally, CASs were identified by correlating the electronic structures of the Fe 3d orbitals with SCR performance. This work clearly shows that Fe³⁺_{Td} sites are CASs in SCR.

MATERIALS AND METHODS

γ -Fe₂O₃, Zn- γ -Fe₂O₃, and Ti- γ -Fe₂O₃ were prepared by coprecipitation (see details in the Supporting Information). According to the X-ray fluorescence (XRF) spectra (Bruker-AXS S4 Explorer) (Table S1), the molar ratios of Ti to Fe and Zn to Fe are 1:3.3 and 1:3.0, respectively, from which Ti- γ -

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Fe_2O_3 and $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$ can be roughly formulated as $(\text{Ti}_{0.7}\text{Fe}_{2.3})_{1-\delta}\text{O}_4$ ($0 < \delta < 0.18$) and $(\text{Zn}_{0.8}\text{Fe}_{2.2})_{1-\delta'}\text{O}_4$ ($0 < \delta' < 0.03$), respectively.

Catalytic evaluation, material characterization, and theoretical calculations^{15–19} are provided in detail in the Supporting Information.

RESULTS AND DISCUSSION

A high GHSV of 300000 h^{-1} was adopted to differentiate the catalytic activities of $\gamma\text{-Fe}_2\text{O}_3$, $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$, and $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$ in SCR. Figure 1 and Figure S1 show the NO conversions (X_{NO})

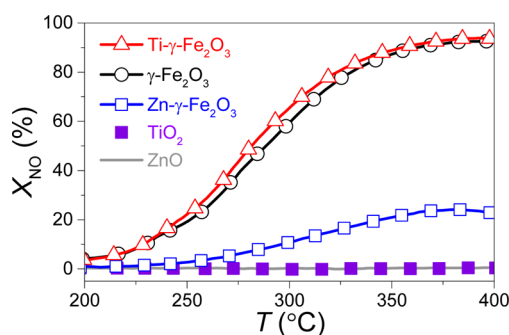


Figure 1. NO conversions (X_{NO}) as a function of temperature (T) over $\gamma\text{-Fe}_2\text{O}_3$, $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$, and $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$ together with TiO_2 and ZnO . Reaction conditions: 500 ppm of NO, 500 ppm of NH_3 , 3 vol % O_2 , balance N_2 , and GHSV of 300000 h^{-1} .

and the selectivity toward N_2 as a function of reaction temperature, respectively. Considering the Zn^{2+} and Ti^{4+} cations cannot usually undergo redox transitions and both TiO_2 and ZnO are inactive in SCR, substituting Ti^{4+} or Zn^{2+} in $\gamma\text{-Fe}_2\text{O}_3$ should not yield extra CASs. As the reaction temperature increases, X_{NO} over pure $\gamma\text{-Fe}_2\text{O}_3$ increases, and X_{NO} reaches $\sim 90\%$ at 400 °C. The specific reaction rates (R) related to surface areas (Table S2) of the catalytic materials are shown in Figure S2. The almost identical X_{NO} (or R) and N_2 selectivity curves can be observed for $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$ except for slight differences in the temperature range of 260–340 °C. In particular, $\gamma\text{-Fe}_2\text{O}_3$ and $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$ almost give the same catalytic activities in the kinetic reaction regime ($X_{\text{NO}} < 15\%$), indicating that Ti doping almost has no influence on the catalytic activity of $\gamma\text{-Fe}_2\text{O}_3$. Similarly, Li et al. reported the SCR activity of $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$ is comparable to that over $\gamma\text{-Fe}_2\text{O}_3$.^{8,14} Hence, the Fe^{3+} sites substituted with catalytically inactive Ti^{4+} in $\gamma\text{-Fe}_2\text{O}_3$ should not be CASs. On the contrary, $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$ gives a SCR activity significantly lower than that of $\gamma\text{-Fe}_2\text{O}_3$. The catalytic activity of $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$ only slightly increases with reaction temperature, and the X_{NO} values are $< 25\%$ in the whole temperature window, similar to the previously reported results.⁸ This demonstrates that inactive Zn^{2+} ions occupy CASs of $\gamma\text{-Fe}_2\text{O}_3$ in SCR; i.e., the Fe^{3+} sites substituted with catalytically inactive Zn^{2+} in $\gamma\text{-Fe}_2\text{O}_3$ are CASs. These results are further confirmed with varying amounts of doped Ti and Zn (Figure S3).

The reaction kinetics of SCR at a low X_{NO} of $< 15\%$ were studied. The related Arrhenius plot in terms of the steady-state reaction rate constant (k) is shown in Figure S4, from which the corresponding apparent activation energies (E_a) can be calculated. The average E_a for $\gamma\text{-Fe}_2\text{O}_3$ is ~ 102 kJ mol^{-1} , equal to that for $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$, indicating that SCR over $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$ and SCR over $\gamma\text{-Fe}_2\text{O}_3$ possibly follow the same reaction

mechanism. The calculated pre-exponential factor (Γ) for $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$ (2.5×10^{12} $\text{L s}^{-1} \text{g}^{-1}$) approaches that (2.2×10^{12} $\text{L s}^{-1} \text{g}^{-1}$) of $\gamma\text{-Fe}_2\text{O}_3$, but it is 5 orders of magnitude larger than that (2.8×10^7 $\text{L s}^{-1} \text{g}^{-1}$) of $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$. The close Γ values imply the same CASs for both $\gamma\text{-Fe}_2\text{O}_3$ and $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$, and the significantly small Γ value for $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$ indicates that the Fe^{3+} sites substituted with Zn^{2+} are CASs of $\gamma\text{-Fe}_2\text{O}_3$.

Figure 2a shows the SXRD patterns. $\gamma\text{-Fe}_2\text{O}_3$ is identified as maghemite with a cubic spinel structure (JPCDS Card No. 39-

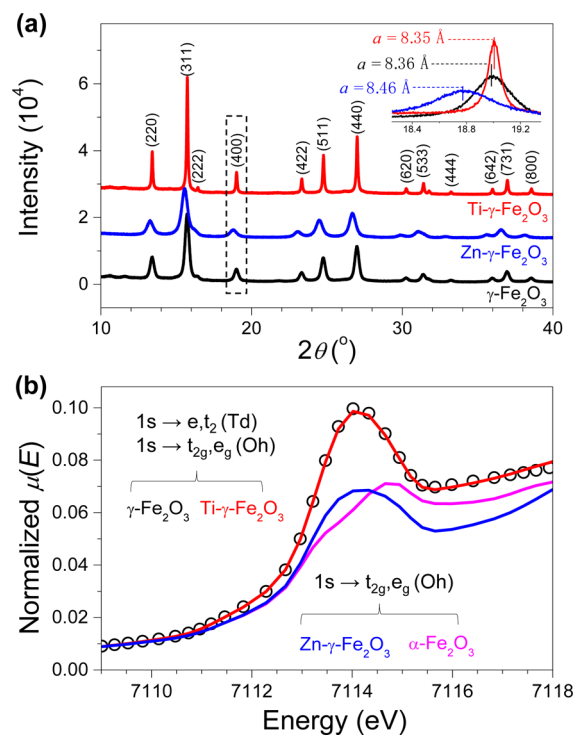


Figure 2. (a) SXRD patterns of $\gamma\text{-Fe}_2\text{O}_3$, $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$, and $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$. (b) Pre-edge XANES spectra of $\gamma\text{-Fe}_2\text{O}_3$ (O), $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$ (red), and $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$ (blue) together with $\alpha\text{-Fe}_2\text{O}_3$ (pink) at the Fe K-edge.

1346). The spinel structure is preserved after substitution with Ti^{4+} or Zn^{2+} , and no TiO_2 and ZnO phase appears in the SXRD pattern of $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$ or $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$, respectively. Note that in comparison to those of $\gamma\text{-Fe}_2\text{O}_3$, all the peaks of $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$ distinctly shift to lower Bragg angles, reflecting the substitution of some Fe^{3+} with Zn^{2+} in $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$ by considering that the ionic radius of $\text{Zn}_{\text{Td}}^{2+}$ (0.60 Å) is larger than that of $\text{Fe}_{\text{Td}}^{3+}$ (0.49 Å).²⁰ Similarly, compared to the peaks of $\gamma\text{-Fe}_2\text{O}_3$, all $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$ peaks shift to slightly larger Bragg angles, demonstrating the replacement of some Fe^{3+} with Ti^{4+} due to the ionic radius of $\text{Ti}_{\text{Oh}}^{4+}$ (0.61 Å) being slightly smaller than that of $\text{Fe}_{\text{Oh}}^{3+}$ (0.65 Å). Furthermore, the related changes in the lattice parameters calculated from the (400) peaks also indicate the successful substitution of Fe^{3+} with Ti^{4+} or Zn^{2+} (inset of Figure 2a).

The immediate structures of doped Ti^{4+} and Zn^{2+} were determined by XANES and EXAFS. The pre-edge features of the Fe K-edge XANES spectra are sensitive to the local geometrical structures of iron centers.²¹ Figure 2b shows the pre-edge XANES spectra of $\gamma\text{-Fe}_2\text{O}_3$, $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$, and $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$ at the Fe K-edge together with a “standard” $\alpha\text{-Fe}_2\text{O}_3$ comprising only octahedral FeO_6 structure. A relatively intense pre-edge peak appears in the XANES spectrum of $\gamma\text{-Fe}_2\text{O}_3$, attributed to the $1s \rightarrow 3d$ (e, t_2) dipole-allowed transition in

$\text{Fe}^{3+}_{\text{Td}}$ centers and the weak $1s \rightarrow 3d$ (t_{2g}, e_g) quadrupole-allowed transition in $\text{Fe}^{3+}_{\text{Oh}}$ centers, but the $1s \rightarrow 3d$ (t_{2g}, e_g) dipole transition is symmetrically forbidden for $\text{Fe}^{3+}_{\text{Oh}}$ centers.²¹ After Ti^{4+} doping, the intensity of the pre-edge peak of $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$ remains unchanged, indicating that Ti^{4+} did not replace $\text{Fe}^{3+}_{\text{Td}}$ sites of $\gamma\text{-Fe}_2\text{O}_3$. According to the SXRD data and the EXAFS spectrum of $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$ at the Ti K-edge (Figure S5), it is convincing to deduce that Ti^{4+} ions occupy $\text{Fe}^{3+}_{\text{Oh}}$ sites of spinel $\gamma\text{-Fe}_2\text{O}_3$.²² Note that the intensity of the pre-edge peak at the Fe K-edge apparently decreases in the XANES spectrum of $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$, and the normalized absorption coefficient (μ) is ~ 0.07 , the same as that of $\alpha\text{-Fe}_2\text{O}_3$ comprising only $\text{Fe}^{3+}_{\text{Oh}}$. This is strong evidence that the Fe^{3+} sites substituted with Zn^{2+} are $\text{Fe}^{3+}_{\text{Td}}$ sites in $\gamma\text{-Fe}_2\text{O}_3$, consistent with the fact that Zn^{2+} ions energetically favorably occupy the tetrahedral sites of spinel structures.²³

EXAFS spectra were applied to further probe the substituted sites with Ti^{4+} and Zn^{2+} (Figures S6 and S7). The Fourier transform (FT) amplitudes of the $\chi(R)$ k^3 -weighted spectra at the K-edge are shown in Figure 3. The $\chi(R)$ k^3 -weighted FT

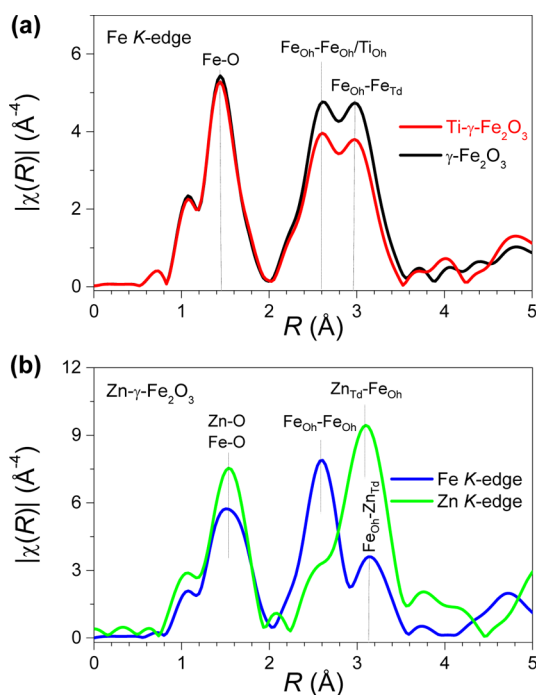


Figure 3. $\chi(R)$ k^3 -weighted FT EXAFS spectra of (a) $\gamma\text{-Fe}_2\text{O}_3$ and $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$ at the Fe K-edge and (b) $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$ at the Fe and Zn K-edges.

EXAFS spectrum of $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$ at the Fe K-edge is similar to that of $\gamma\text{-Fe}_2\text{O}_3$, except that the decreases in intensity in the second and third shells of $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$ are observed. The first shell can be attributed to the contribution from the Fe–O bonds with a bond length of ~ 1.97 Å (Table S3),²² in line with the average Fe–O bond length (~ 2.00 Å) in $\text{Fe}^{3+}_{\text{Td}}$ and $\text{Fe}^{3+}_{\text{Oh}}$ sites in $\gamma\text{-Fe}_2\text{O}_3$.²⁴ The second shell is ascribed to the $\text{Fe}_{\text{Oh}}\text{-Fe}_{\text{Oh}}$ and $\text{Fe}_{\text{Oh}}\text{-Ti}_{\text{Oh}}$ interatomic distances, and the third shell should be assigned to the $\text{Fe}_{\text{Td}}\text{-Fe}_{\text{Oh}}$ interatomic distance, which also covers the scattering paths of Fe_{Td} to Ti_{Oh} for $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$. These amplitudes at the second and third shells are weaker than the corresponding amplitude of $\gamma\text{-Fe}_2\text{O}_3$, largely because the atomic number of Ti is smaller than that of Fe.

The three similar shells appear in the FT EXAFS spectrum at the Fe K-edge of $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$. The attributions of the first and second shells are shown in Figure 3b, and the amplitude of the third shell is much weaker than the corresponding amplitude of $\gamma\text{-Fe}_2\text{O}_3$, attributed to the interatomic distance between Fe_{Oh} and Zn in the tetrahedral sites. To further substantiate the Zn^{2+} position, we also recorded the FT EXAFS spectrum of $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$ at the Zn K-edge (also shown in Figure 3b). The absence of ~ 3.0 Å interatomic distances demonstrates that Zn^{2+} ions are not located at the octahedral sites of spinel.²² A combination of the SXRD data with the XANES and EXAFS spectra reveals that Ti^{4+} and Zn^{2+} substituted for the octahedral and tetrahedral sites of spinel $\gamma\text{-Fe}_2\text{O}_3$, respectively.

As shown in Figure 1, the catalytic performance of $\gamma\text{-Fe}_2\text{O}_3$ remains almost preserved after $\text{Fe}^{3+}_{\text{Oh}}$ sites are substituted with catalytically inactive Ti^{4+} , but $\gamma\text{-Fe}_2\text{O}_3$ loses activity after $\text{Fe}^{3+}_{\text{Td}}$ sites are substituted with Zn^{2+} , indicating that $\text{Fe}^{3+}_{\text{Td}}$ sites are CASs in SCR rather than $\text{Fe}^{3+}_{\text{Oh}}$ sites. Redox cycles are often involved in SCR regardless of the reaction mechanisms (Eley–Rideal and Langmuir–Hinshelwood models),^{14,25} and the $\text{Fe}^{3+} \leftrightarrow \text{Fe}^{2+}$ redox cycles occurring during the SCR process are intimately associated with the catalytic activity. Generally, Fe^{3+} sites are responsible for NH_3 oxidation in SCR,^{5–8} concomitant with $\text{Fe}^{3+} \rightarrow \text{Fe}^{2+}$ transfer, and the subsequently produced Fe^{2+} sites are oxidized to Fe^{3+} by O_2 to finish the redox cycle. The $\text{Fe}^{3+} \rightarrow \text{Fe}^{2+}$ transfer is relatively easy because Fe^{3+} has strong oxidative ability, and the $\text{Fe}^{2+} \rightarrow \text{Fe}^{3+}$ transfer depends on coordinative configurations of iron ions. Ponec et al. used the isotopic exchange reaction of oxygen to investigate the interactions of Fe_2O_3 (it should be $\alpha\text{-Fe}_2\text{O}_3$ with only $\text{Fe}^{3+}_{\text{Oh}}$ because of an extremely high melting point) with $^{18}\text{O}_2$ and found that the E_a value for $^{18}\text{O}_2(\text{g}) + 2^{16}\text{O}_s \rightarrow ^{16}\text{O}_2(\text{g}) + 2^{18}\text{O}_s$ [where $\text{O}_2(\text{g})$ and O_s represent gaseous O_2 and surface lattice oxygen of Fe_2O_3 , respectively] is very high, i.e., ~ 130 kJ mol⁻¹ (~ 1.4 eV per $\text{Fe}^{3+}_{\text{Oh}}$) even at 400–500 °C, although the Fe–O bond energy is only 65.7 kJ mol⁻¹.²⁶ This suggests that the activation energy to realize $\text{Fe}^{2+}_{\text{Oh}} \rightarrow \text{Fe}^{3+}_{\text{Oh}}$ transfer should be higher than the activation energy (~ 1.1 eV) of SCR over $\gamma\text{-Fe}_2\text{O}_3$ in Figure S4.

To investigate the sustainability of the $\text{Fe}^{3+} \leftrightarrow \text{Fe}^{2+}$ redox cycles at the tetrahedral sites, we collected Fe L_3 -edge X-ray absorption spectra (Figure 4a). According to the O_h symmetry of Fe and the crystal-field theory, two peaks in the Fe L_3 -edge X-ray absorption spectra of $\gamma\text{-Fe}_2\text{O}_3$, $\text{Ti-}\gamma\text{-Fe}_2\text{O}_3$, and $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$ can be assigned to the $2p_{3/2} \rightarrow 3d$ (t_{2g}, e_g) transitions with an octahedral ligand-field splitting energy (Δ_{Oh}) of ~ 1.6 eV.²² The high-spin electron configurations of $\text{Fe}^{3+}_{\text{Oh}}$ and $\text{Fe}^{2+}_{\text{Oh}}$ are $t_{2g}^3e_g^2$ and $t_{2g}^4e_g^2$, respectively, and thus, the unoccupied states of the t_{2g} orbitals of $\text{Fe}^{3+}_{\text{Oh}}$ outnumber those of $\text{Fe}^{2+}_{\text{Oh}}$. As a consequence, the unoccupied states of the t_{2g} orbitals are positively proportional to their corresponding intensity in the Fe L_3 -edge X-ray absorption spectra. Hence, a ratio (λ) of t_{2g} to e_g in intensity in the Fe L_3 -edge X-ray absorption spectrum can be used to determine the oxidation states of Fe_{Oh} .²⁷ The λ value becomes smaller after Ti^{4+} substitutes for $\text{Fe}^{3+}_{\text{Oh}}$ in $\gamma\text{-Fe}_2\text{O}_3$, concomitant with the $\text{Fe}^{3+} \rightarrow \text{Fe}^{2+}$ transfer in another octahedral site rather than that in the tetrahedral site. This clearly demonstrates that $\text{Fe}^{2+}(t_{2g}^4e_g^2)$ is more stable in the octahedral site than in the tetrahedral site; i.e., the propensity of $\text{Fe}^{2+}_{\text{Td}}$ to be oxidized to $\text{Fe}^{3+}_{\text{Td}}$ should be higher than that of $\text{Fe}^{2+}_{\text{Oh}}$ to $\text{Fe}^{3+}_{\text{Oh}}$. The λ values for $\text{Zn-}\gamma\text{-Fe}_2\text{O}_3$ and $\gamma\text{-Fe}_2\text{O}_3$ are equal to each other, indicating that the $\text{Fe}^{3+}_{\text{Oh}}(t_{2g}^3e_g^2)$ electron configuration with a Δ_{Oh} value of ~ 1.6

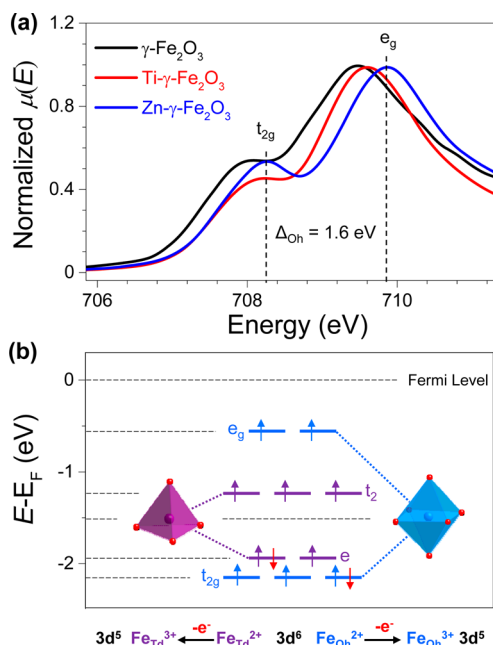


Figure 4. (a) Fe L_3 -edge X-ray absorption spectra of γ - Fe_2O_3 , Ti- γ - Fe_2O_3 , and Zn- γ - Fe_2O_3 . (b) Electron configurations of the tetrahedral and octahedral sites in γ - Fe_2O_3 . The inset shows the FeO_4 tetrahedron (purple) and FeO_6 octahedron (blue). Red and blue spheres represent O and Fe, respectively.

eV remains preserved after Zn^{2+} replaces $\text{Fe}^{3+}_{\text{Td}}$ (Figure 4a). Therefore, according to the crystal-field theory, a tetrahedral ligand-field splitting energy (Δ_{Td}) for Fe_{Td} in γ - Fe_2O_3 can be calculated to be ~ 0.7 eV ($\Delta_{\text{Td}} = \frac{4}{9}\Delta_{\text{Oh}} = \frac{4}{9} \times 1.6$ eV).

The Fe 3d electronic configurations of γ - Fe_2O_3 are plotted in Figure 4b on the basis of the analyses of the Fe L_3 -edge X-ray absorption spectra in Figure 4a described above. We further scale the energy of these orbitals by using the DOS of surface γ - $\text{Fe}_2\text{O}_3(110)$ planes [the exposed (110) surfaces were observed by transmission electron microscopy in Figure S8], which were calculated by DFT (Figure S9). The Fe 3d centroid is calculated to be approximately -1.52 eV, and both $\text{Fe}^{3+}_{\text{Oh}}$ and $\text{Fe}^{3+}_{\text{Td}}$ are high-spin forms because Δ_{Oh} and Δ_{Td} are smaller than the electron pairing energy.²⁸ In Figure 4b, the $\text{Fe}^{2+}_{\text{Td}} \rightarrow \text{Fe}^{3+}_{\text{Td}}$ transfer by releasing one electron from the $\text{Fe}^{2+}_{\text{Td}}$ e orbital is relatively easier than the $\text{Fe}^{2+}_{\text{Oh}} \rightarrow \text{Fe}^{3+}_{\text{Oh}}$ transfer, indicating that $\text{Fe}^{2+}_{\text{Td}}$ is easily oxidized by O_2 to $\text{Fe}^{3+}_{\text{Td}}$. These results agree fairly with the fact that spinel iron oxides often have inverse spinel structures with Fe^{3+} occupying the tetrahedral sites, consistent with the findings that $\text{Fe}^{3+}_{\text{Td}}$ ions are so active that they preferentially underlie the subsurface layers.¹³ Therefore, this work gives a simple and straightforward explanation for why $\text{Fe}^{3+}_{\text{Td}}$ ions of γ - Fe_2O_3 act as CAS in SCR, and it provides a strategy for enhancing the thermal stability to γ - Fe_2O_3 without sacrificing its catalytic performance (Figure S10). Such a redox cycle in γ - Fe_2O_3 may not be unique, because the oxides of Co, Mn, and Ni have similar spinel structures that are often subjected to oxidation–reduction equilibria.²⁹ Further work is necessary to precisely describe mechanisms of the $\text{Fe}^{3+} \leftrightarrow \text{Fe}^{2+}$ redox cycles in practical SCR.

■ ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.estlett.7b00124.

Some related tables and figures (PDF)

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Notes

The authors declare no competing financial interest.

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